Crown ether containing polyelectrolyte multilayer 1 membranes for lithium recovery 2 Mohammad Kazemabad^{a,*}, Arne Verliefde^a, Emile R. Cornelissen^{a,b}, 3 Arnout D'Haese^a 4 ^aParticle and Interfacial Technologies group, Faculty of Bioscience Engineering, Ghent 5 University, Coupure Links 653, B-9000, Ghent, Belgium 6 ^bKWR Water Cycle Research Institute, Groningenhaven 7, 3433 PE, Nieuwegein, The 7 Netherlands 8

9 Abstract

Achieving solute selectivity has always been a goal of membrane development studies. The continuing growth of global consumption of scarce metals by different industries has put a strain on traditional sources of these species. Achieving cation selectivity in membranes, especially among monovalent cations, is a major step in introducing alternative sources for scarce metals such as lithium.

Polyelectrolyte multilayer membranes (PEMMs) are a novel class of 16 membranes, offering great potentials in monovalent/bivalent ion selectiv-17 ity. On the other hand, crown ethers are a well-studied family of macro-18 cyclic ligands capable of forming stable complexes with cations. In the 19 current study, for the first time, we report on a PEMM nanofiltration mem-20 brane with crown ether moieties embedded in its structure for the goal of 21 achieving monovalent salt selectivity. The crown ether 15-crown-5 was suc-22 cessfully incorporated in the polycation polyethylenimine (PEI), which was 23 then used as the polycation in PEMM formation through layer by layer de-24 position. Both the synthesized polymer and the polyelectrolyte multilayer 25 (PEM) were characterized and the performance of the resulting membrane 26

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was studied. It was determined that crown ether containing polymer forms 27 more stable complexes with lithium than potassium. This was explained 28 by the limitation put on 2:1 potassium-crownether complexes by steric hin-29 drance from polymer chain. The manufactured membranes showed Li/K 30 selectivity for a period of around 90 minutes, after which the crown ethers 31 became saturated and selectivity was lost. The modified membranes became 32 non selective after this point, but possessed high salt rejection potential. 33 Keywords: polyelectrolyte multilayer membranes, layer by layer 34 deposition (LBL), Crownethers, Lithium recovery, Selective separation 35

36 1. Introduction

Application of lithium in Li-ion batteries and the continuous shift from 37 fossil-based energies towards electrical energy, has put a high pressure on 38 conventional lithium resources. Based on the conservative estimations, cur-39 rent lithium reserves are projected to run out by 2080 [1]. Alternative 40 lithium sources such as sea water, industrial and mining waste waters, RO 41 brines and spent lithium batteries can significantly lower the toll on the 42 current reserves [2]. For example, assuming a recovery efficiency of 80%. 43 more than 60% of the worldwide lithium consumption can be recovered 44 from RO desalination brine[3]. Lately, tremendous amount of effort has 45 been directed at the search for novel lithium sources and recovery methods 46 by both academia and industry. Diallo and co-workers [3] have reviewed 47 prospects of critical metal recoveries from seawater. Meshram et al. [4] 48 have investigated traditional and novel methods of lithium recovery from 49 primary and secondary sources. Although recovery from alternate sources 50 is still very limited in an industrial scale, a number of small scale plants 51 aimed at lithium harvesting from the seawater exist. For example the In-52 stitute of ocean energy at Saga University in Japan prepared a lab scale 53 set-up for lithium extraction from seawater based on MnO_2 adsorbers [4, 5]. 54 In another attempt, Korea institute of geoscience and mineral resources 55 (KIGAM) partnered with steel making industry POSCO in 2010 to launch 56 a seawater lithium harvesting plant with the aim of reaching 30 tonnes of 57 production over a course of four years. [6] 58

The greatest obstacles of lithium harvesting from alternative sources are the low concentration of lithium, and the presence of other alkali metals. Therefore development of a separation process capable of differentiat-

ing among different monovalent cations is of great importance. Absorbents 62 of ion sieve type (such as mangenese oxide $H_{1.6}Mn_{1.6}O_4$ and spinel-type 63 $\lambda - MnO_2$) are one of the most promising solutions to the selectivity ob-64 stacle, with the later capable of concentrating lithium up to 400 times from 65 a brine with high concentrations of sodium [4, 7]. However the purity of 66 the extracted lithium can be still low ($\approx 33\%$) and sodium contamination 67 can still be problematic. Crownether based adsorbents are another class of 68 cation selective adsorbents studied. 69

Layer by layer (LBL) deposition of polyelectrolytes is a membrane fab-70 rication method which results in formation of ultra-thin poly electrolyte 71 layers as active layer. Compared to other thin film deposition methods, 72 LBL assembly is extremely cheap, simple and versatile^[8] and provides the 73 possibility of controlling the film properties while offering excellent flexi-74 bility and fine tuning [9]. Therefore polyelectrolyte multilayer membranes 75 (PEMMs) have enjoyed considerable interest in recent years. An overview 76 of the LBL literature and the PEMMs can be found in reviews [10, 11, 12]. 77 Since the main mechanism of ion rejection in PEMMs is based on charge 78 rather than size, LBL assembleed membranes are known to reject divalent 79 ions better than the monovalents [10], however monovalent selectivity re-80 mains challenging. 81

Crownethers are a family of macrocyclic ligands well known for their ability to form stable complexes with cations. Agreement between the crown ether cavity size and the cation size is the main factor affecting the stability of the cation-ligand complex formed. This was originally proposed by Pedersen [13] and later confirmed by X-ray crystallography [14]. However, it only provides a rule of the thumb, leaving out several other factors that can influence and even reverse the order of cation preference[15].

The mechanism of the complex formation in crownethers involves a com-89 plete or partial shedding of the solvation shell of the dissolved ions and 90 subsequent solvation of the formed complex. Therefore the nature of the 91 medium in which complexation is taking place plays a vital role in the ther-92 modynamics of the reaction. It should be mentioned that in our case, the 93 complexation is taking place inside the membrane pores, where an accurate 94 definition of the medium is debated. It is known that physical properties of 95 confined water in nanopores and especially its dielectric constant is differ-96 ent from that of the bulk water [16, 17]. Moreover, hydration shells of the 97 cations are prone to rearrangement or even breakage inside the nanopores 98 [18]. It can thus be expected that crownethers confined in nanopores can 99 act different than when they are free in the bulk. 100

Selectivity of the crownethers is also affected by the ring flexibility [15, 101 19]. Modifying the ring rigidity is regularly used to increase selectivity[19, 102 20]. The nature of the chemical groups in immediate vicinity to the ether 103 groups also have a profound effect on the strength of the interaction. The 104 electron density of the adjacent oxygen atoms and thus the binding capacity 105 of the ligand are affected by attached groups. For example, it is shown that 106 K/Na selectivity can be reversed in crownethers with electron releasing or 107 electron attracting substituents [15]. It is also shown that extremely stable 108 complexes can be formed when there are anionic groups attached to the 109 crownether[21]. Therefore, the cation-crownether complexes in the PEMMs, 110 where anionic groups from the polyanion chain are present, are expected to 111 be stronger. It's been reported that the response of different cations to the 112 presence of the anionic groups is unpredictable and generalization is not 113 possible [15]. 114



Based on the relative size of the cation and the cavity, the formed com-

plex can be of different stoichiometries. The 2:1 sandwich chemistry, in 116 which the cation is present on top of one crownether and is sandwiched 117 from the other side by another, provides the complex formation possibility 118 for larger cations [22, 15]. For example it is shown that in solid state, the 119 potassium complex in benzo-15-crown-5 (the precursor used in this work) 120 is of the sandwich type[23]. In some cases existence of such higher order 121 chemistries compromises the selectivity potential of crownethers. Forma-122 tion of this type of complexes can be limited by adding bulky subunits 123 to the ether ring, blocking the formation of sandwich complexes by steric 124 hindrance [19, 24, 25, 26]. In a similar manner, polymerization of crownethers 125 can affect their potential for forming higher order complexes. It is known 126 that the order of cation preference changes in poly-crownethers with respect 127 to their monomers. For example, Kimura *et al.* found that polymers of 128 18-crown-6 and 15-crown-5 have higher binding constants for cations that 129 tend to form 2:1 complexes than their respective monomers, whereas binding 130 constants of the smaller cations which form 1:1 complex remain more or less 131 unchanged [27]. We expect that incorporation of the crownethers in highly 132 branched PEI polymer in this work, introduces high steric hindrances and 133 seriously limits the 2:1 complexing ability. 134

Considering all the above mentioned factors, it is inherent that the order of cation preference in the polymer synthesized in this project can't be predicted from its precursor crownethers. To the best knowledge of the writers, this is the first time this modification route is being taken for PEI as a polymer for membrane fabrication and thus no prior data is available on the complex strength of different cations and the crownether.

¹⁴¹ Crownethers have been used in several studies on membranes. How-¹⁴² ever, in most cases, crownethers serve as the carrier in carrier mediated

transports. To the best knowledge of the writers, this is the first report on 143 usage of crownethers in a pressure driven nanofiltration membrane. Liq-144 uid membranes (LMs), a family of membranes where a liquid organic phase 145 separates two aquatic phases, build up a substantial part of the literature 146 on crownethers [28, 29, 30, 31]. The most important problems of this class 147 of membranes are leaching of the organic solvent or carrier and contamina-148 tion of the aquatic phases due to the limited solubilities [32, 33, 34]. As 149 a close relative to LMs, plasticized polymeric membranes reportedly house 150 semi-liquid pockets in between their crystalline parts [35]. Crownethers have 151 been used as fixed carriers in plasticized cellulose triacetate (CTA) for gold. 152 silver, copper [34] and cesium [36, 37] ion transport. 153

Crown ethers have also been regularly used in ion exchange resins and 154 membranes. The group of Bartsch developed several polycrownethers with 155 acidic side arms to be used as ion exchange resins to achieve alkali metal 156 [38, 39] and lead (II) [40] selectivity. Favre-Reguillon et al. [41] immobilized 157 a number of crown ethers on polyure thans and studied the effect of polymer 158 chain rigidity on the complexation. More recently, Tas and co-workers in-159 corporated 18-crown-6 in Poly arylene ether ketone (PAEK) backbone and 160 blended it with sulfonated poly ether ether ketone (SPEEK) to prepare 161 monovalent cation selective membranes which they tested under diffusion 162 dialysis [42]. 163

Crownethers can be added to the membrane by either blending or covalent bonding to the matrix. Blending crownethers with the polymer solution during phase inversion or interfacial crystallization is one of the most common methods of embedding crown ethers in membrane structure. In the resulting membranes, crown ethers are trapped in the polymer matrix and therefore more immune to leaching [43]. Major disadvantages of this method

are the promotion of structural defects and the possibility of demixing and 170 uneven distribution [35]. Dulyea and co-workers prepared phase inverted 171 membranes by blending a crownether containing polymer with an otherwise 172 impermeable polymer. The resulted membranes showed a Na/K selectivity 173 of 1.8.1. However, the membrane structure was disrupted at high crownether 174 concentrations and defects in membrane structure even at low concentrations 175 were reported [33]. Therefore, embedding of crownethers through covalent 176 bonds in polymers is believed to yield more stable membranes. An overview 177 of different methods to incorporate crownethers in polymers is given by 178 Alexandratos and Stine [43]. 179

Careful consideration must be given in selection of the crown ether moi-180 ety. Due to the close resemblance of the ionic radii and chemical properties of 181 alkali metals, most of the crown ethers form complexes (with varying degrees 182 of stability) with multiple ions. An optimal performance of the membrane 183 can be achieved by selecting a crown ether which has a fundamentally differ-184 ent reaction to the desired target ion. In our case, this translates in forming 185 complexes with all alkali metals except lithium. The complex formation 186 constants from table 1 confirm such a property for 15-crown-5. Lithium-187 15-crown-5 complexes are unstable in water [44], whereas potassium and 188 sodium are capable of forming stable complexes [45, 46, 47, 48]. 189

190 1.1. Goal of the study

PEMMs possess a natural monovalent/bivalent selectivity which makes them suitable for development of selective membranes. Monovalent selectivity can be introduced to these membranes by inclusion of crown ethers. Here, we report the covalent attachment of the crown ethers as pendant groups to the poly cation chain and subsequent LBL deposition of the modified

Crownether	Cation	$\rm Log~K_a$	Medium	Method	reference
15-crown-5	Li	≈ 0	H_2O	NMR	[44]
	Na	0.44	H_2O	NMR	[47]
	Na	0.70	H_2O	Calorimetry	[45]
	Κ	0.74	H_2O	Calorimetry	[45]
benzo-15-crown-5	Na	0.40	H_2O	Electrophoresis	[48]
	Κ	0.38	H_2O	Electrophoresis	[48]
	K	0.38	H ₂ O	Calorimetry	[46]

Table 1: Thermodynamic data of complexes of cations with 15-crown-5 and benzo-15-crown-5.

polycation and polyanion to form a lithium selective nanofiltration membrane. It was crucial to investigate membrane stability at high salinities
since PEMMs are known to be susceptible to high salinity and proposed
alternative lithium sources such as RO brines are highly concentrated.

One of the important differences of the membrane prepared in this study 200 and the majority of the crown ether containing membranes such as LMs is 201 the nature of the matrix. The organic liquid phase of LMs are virtually 202 impermeable to water and ions and crown ethers offer the only ion perme-203 ation pathway. Whereas normal PEMMs are permeable to water and ionic 204 species. In our case, ion exchange sites intrinsic to the polyelectrolytes are 205 much more abundant than crownethers. Therefore "ion hopping" can be ex-206 pected to take place through these sites. Incorporation of the crownethers 207 in the PEMM structure introduces a number of complexation sites in the 208 membrane. Strong complexes formed in these sites can act as an Ion trap, 209 hindering the transport of the cation across the membrane. The differences 210

in the complex strength between different cations is reflected in the selec-211 tivity. Based on the complexation strength, trapped ions need to overcome 212 a certain energy barrier to move past the membrane. Therefore we expect 213 the membrane to hinder the transport of complexing cations. It can thus 214 be expected that the membrane will reject the complex forming ions more, 215 and cations with the less stable complexes will be more abundant in the 216 permeate. Based on the thermodynamic data (Table 1), we expect a reduc-217 tion in the transport rate of all alkali metals in the crown ether containing 218 membranes. However, this reduction will be more severe in the case of 219 sodium and potassium in comparison to lithium, resulting in a lithium rich 220 permeate. 221

222 2. Materials and methods

223 2.1. Synthesis and characterization of crownether incorporated polycation

224 2.1.1. 4'formylbenzo 15 crown-5

4'Formylbenzo 15-crown-5 was synthesized by a ring closure reaction 225 between a linear ether and 3,4-dihydroxybenzaldehyde based on the method 226 of Ungaro et al. [49] with slight modifications. A detailed explanation of 227 the synthesis procedure can be found in the supporting information. In 228 total, 3.9 grams of crownether crystals were purified, corresponding to a 229 yield of around 20%. The low yield of the procedure is attributed to two 230 factors: undesired reactions and loss of product during purification steps. 231 Of these, the former is responsible for almost half of the loss. Possible 232 side reactions include monosubstitued reaction of the hydroxyl groups of 233 dihydroxybenzaldehyde resulting in an open ring and the association of the 234 benzaldehyde to two different ether chains. 235



Figure 1: synthesis route for 4'formylbenzo-15crown5.

The nature and purity of the resulting crystal was studied by NMR spectra. HNMR was performed at 400MHz with a Bruker Avance III Nanobay 400 MHz spectrometer. Tetramethylsilane (TMS) was used as reference compound. It was determined that the final crystals had a purity of about 95%. The main remaining impurity was unreacted linear ether which was not removed during the purification, with HNMR peaks between 3.6 and 3.8 ppm.

243 2.1.2. Incorporation of crown ether in PEI

The synthesized formylbenzo 15 crown-5 was incorporated in the Polyethylen-244 imine (PEI) structure by reductive amination. The protocol was adopted 245 from the work of Kasprzak et al. who added chemically similar p-formylbenzoic 246 acid to PEI [50]. The PEI solution (50 wt% in water, highly branched, M_w 247 750kDa, M_n 60kDa, Sigma-Aldrich) was first dried using a rotavapor un-248 der vacuum. 50 ml of methanol was then charged with 2.5 g of dry PEI, 249 15 mg (0.25 mmol) of acetic acid and 375 mg (1.25 mmol) of synthesized 250 4'-formylbenzo 15crown-5. The mixture was refluxed for one hour and then 251 left to stir at room temperature for 24 hours. 190 mg (5.0 mmol) of NaBH₄ 252 was then added to the mixture as the reducing agent and stirring continued 253 for another 4 hours at room temperature. 254



Figure 2: crownether incorporated polyetherimine.

255 2.1.3. Polymer purification and characterization

To remove the unbound crownethers and any other byproducts from 256 modified PEI, the solution was extracted three times with ethyl acetate, 257 leaving PEI in the aqueous phase. The resulting solution was then dialysed 258 using DI water for two days. A Spectra/Por[®] regenerated cellulose dialysis 259 membrane kit with MWCO of 12-14kDa from Spectrum labs (US) was used 260 for this procedure. HNMR spectra (Fig. 3) validated the presence of crown 261 ethers in the polymer structure. Peak were assigned to hydrogens with the 262 help of ACDLabs's 1D NMR Processor software (ACD labs, Canada) (refer 263 to Table.2). From these data, it was determined that crownether has been 264 added in a ratio of one molecule for every 44 amine groups, corresponding 265 to $0.53 \frac{mmol\ crownether}{g\ of\ polymer}$ (for a detailed discussion on calculation of this ratio 266

refer to Appendix B in the supporting material). The stoichiometric ratios of reactants have a direct effect on the ratio of crown ether/PEI monomer in the final product, which in turn affects the performance of the synthesized membranes. However, only two recipes were tested and it was decided that an optimization of the crown ether/monomer ratio is out of the scope of the current study.



Figure 3: HNMR spectrum of crownether corporated polyetherimine in D₂O. please refer to 2 for list of assigned peaks. X axis represents chemical shift (ppm) and Y axis represents normalized intensity ($\times 10^{-3}$)

273 2.2. Validation of PEM formation and evaluation of layer thickness via 274 QCM-D

Highly branched polyethylenimine (PEI) was used as the polycation on
this study. Poly styrenesulfonate (PSS), arguably the most common polyanion, was chosen to be paired with PEI. To validate the formation of the PEM,
quartz crystal microbalance with dissipation (QCM-D) was used. The QCMD technique can determine the mass and physical properties of the film de-

Atom number	H shift (ppm)
26,29,27,30	1.905
$23,\!25,\!28$	2.684
22	2.807
$12,\!14,\!10,\!11$	2.897
$20,\!13,\!15$	3.079
21	3.327
8	3.679
9	3.733
19	7.027
$17,\!16$	7.055

Table 2: Assigned peaks to the NMR spectrum

posited on the surface of the sensor by measuring the change of resonating frequency of a quartz crystal resonator. Measurements were performed on a Q-Sense E4 from Biolin scientific (Sweden), using silica coated quartz crystals with a fundamental frequency of 5Mhz (Biolin scientific-Sweden). A Rheodyne MX seriesII[™] actuated valve from Idex health and science (US) was used as liquid handler for switching between different streams at the designated times.

The procedure of PE coating on the sensors was in close resemblance to the dip coating procedure used (refer to section 2.4). PSS (M_w 70kDa, Sigma-Aldrich) was used as polyanion. A 0.4wt% PE solution in 0.5M NaCl was pumped at 0.1 ml/min for 15 min, followed by the rinsing steps at 0.5M NaCl in DI water. The duration and flow rate of the rinsing step was also 15 min and 0.1 ml/min respectively. Raw data were processed with the help of Qsense-Dfind software provided by Biolin Scientific. A density of 1.2 gr/cm³ was assumed for the polyelectrolyte layers in accordance with the literature [51]. Each set of measurements was performed on four identical sets of sensors and results were averaged.

297 2.3. PEM stability

To assess the stability of the resulting PEM at high salinity levels, a 298 test with QCM was designed. Stability of four set of PEMMs (modified and 299 unmodified polycation, PEMMs ending in positively or negatively charged 300 layers) was studied. Silica coated quartz crystals were coated with either 301 four layers (2 bilayers) or five layers (2.5 bilayers) of PEs. This step is marked 302 in the Fig. 6 in supporting materials as PE coating step. At the end of the 303 coating step, the sensors were rinsed with Milli-Q water to obtain a base-304 line. This is necessary since a change in the salinity results in change of the 305 multilayer swelling and water content. The water content of the membranes 306 are reflected in the QCM results, rendering the stability validation by com-307 parison of membrane thicknesses in saline solutions impossible. Comparing 308 the signals at Milli-Q water eliminates this effect. The multilayer was then 309 put in contact with a saline solution of NaCl with varying concentration, 310 starting from 1 M and raised stepwise in 0.25M steps until reaching 4.75M. 311 Higher concentrations were not pursued due to the risk of NaCl crystalliza-312 tion inside the instrument. Adequate time was given to the PE multilayer 313 to reach a constant frequency $(\Delta f/t < 0.1 Hz/min)$. After this the solution 314 was changed to Milli-Q to quantify mass loss from the PEM. These steps are 315 identifiable in Fig. 5 and Fig.6 in supporting material with their respective 316 NaCl concentrations. 317

318 2.4. Membrane synthesis via LBL assembly

Nanofiltration membrane coupons were prepared via dip coating on a 319 Polyacrylonitrile (PAN) support layer. A PAN ultrafiltration membrane 320 from Synder filtration (US) with a MWCO of 30kDa was hydrolyzed by 321 immersion in a solution of NaOH 1.5M at 45°C for 1.5 hours. Membrane 322 coupons (with both sides exposed) were then immersed in subsequent solu-323 tions of 0.4 wt% PE in the presence of 0.5M of NaCl as supporting electrolyte 324 for 15 minutes. Rinsing was also performed using a 0.5M NaCl solution 325 for 15 minute periods. A complete cycle of bilayer deposition consisted of 326 dipping the membranes in solutions of polycation, rinsing, polyanion and 327 rinsing again. This cycle was repeated three times to obtain 6 layers (three 328 bilayers) of poly electrolytes. The pH of the PSS solution was not adjusted, 329 whereas the pH of the PEI was always adjusted to 9 for both modified and 330 unmodified cases. After coating, the coupons were stored in DI water for at 331 least 48 hours until use. 332

333 2.5. Filtration experiments

Dead-end filtration experiments were carried out to characterize the 334 manufactured membranes. For this purpose, a 300 ml SterlitecTM HP4750 335 stirred cell connected to a nitrogen pressurized vessel was used. The experi-336 ments typically lasted for at least six hours, however, at least one experiment 337 of each data set was continued for up to 48 hours to verify the stability of 338 the results. The permeate was sampled more frequently at the beginning 339 until steady state was reached, after which sampling frequency was reduced. 340 Transmembrane pressure was set at five bars. The filtration vessel was kept 341 under constant stirring at room temperature for the duration of the experi-342 ment. 343

Single salt experiments using LiCl, NaCl and KCl were performed at 344 a feed cation concentration of 200 ppm. Mixed salt experiments were de-345 signed to investigate the effect of cation competition. It is known that the 346 order of cation preference of crownethers can be different in competitive 347 mode than the single cation mode [38]. A solution of LiCl and KCl with 348 a cation concentration of 100 ppm each, was used as feed for mixed salt 349 experiments. Since the membranes were prepared in presence of NaCl, they 350 always contained some NaCl prior to first use. This made an accurate mass 351 balance impossible and it was decided to do not include NaCl in the mixed 352 salt experiments. In between the experiments, membrane coupons needed 353 to be treated to regenerate the crownether sites. In most of the studies on 354 absorbents and ion exchange resins this is done by acid treatment [42, 52]. 355 However PEMMs are unstable at extreme pHs and it was decided to re-356 generate membranes by saturating them with Na ions. This was done by 357 filtering a NaCl 200 ppm solution (similar to single salt NaCl experiments). 358 Membrane coupons which were being used for the first time are designated 359 as 'virgin' whereas all non specified specimen are regenerated coupons. 360

361 3. Results and discussion

362 3.1. PEM formation

Formation of a stable polyelectrolyte multilayer was confirmed by QCM-D experiments in the case of both modified and unmodified PEI. Furthermore reaching a constant frequency ($\Delta f/t < 0.1 Hz/min$) before 15 minute confirmed the suitability of the time span selected for adsorption and rinsing steps (refer to section 2.2). The evolution of layer thickness was linear in the case of PEMs from unmodified PEI. This is in agreement with the data reported in literature [51]. Such a linear growth regime points to a low
degree of intermixing between PE layers [53].

The evolution of PEM thickness was more complex in the case of modi-371 fied PEI/PSS pair (Fig. 5). Instead of an increase in thickness upon addition 372 of a PSS layer, the PEM lost part of its thickness (for example notice lower 373 thickness of the second PSS layer in comparison to the second PEI layer in 374 Fig. 5). This can be explained by lower degree of swelling of the membranes 375 with PSS capping layers. It should be noted that the QCM-D method can 376 not differentiate between the poly electrolyte mass and the water content 377 of the PEM. Fig. 4 shows frequency shift and dissipation evolution during 378 a typical coating procedure of the modified PEI/PSS multilayer. The dissi-379 pation value is consistently lower after PSS rinsing steps than PEI rinsing 380 steps. Lower dissipation point to more rigid films [54] which in case of PEMs 381 are usually result of lower swelling and less looped polyelectrolytes. It can 382 thus be concluded that after addition of PSS layers, polymer films loose 383 their water content and become less thick and more rigid. 384

Multilayers formed from modified PEI were consistently thicker in com-385 parison to their unmodified counterparts. One of the possible explanation 386 of this increased thickness can be the differences in PEM structure such as 387 presence of polymer chain loops and PEM water content and swelling. The 388 ratio of deprotonated amines to all amines in the modified PEIs is most 389 likely to be different from the unmodified counterpart. Another possible 390 explanation comes from the steric hindrance caused by relatively spacious 391 crown ether molecules. Some of the positively charged sites on PEI can 392 become inaccessible to the polyanion by this steric hindrance. Crown ethers 393 limit approaching of the sulfonate groups on PSS to the nearby protonated 394 amines. This reduces the apparent charge density of the PEI chain and 395

³⁹⁶ increases the amount of PEI needed to reach equilibrium and overcompen-

³⁹⁷ sate. A definitive explanation of the nature of the increase in thickness can

398 only come from a more thorough characterization of membranes in future studies.



Figure 4: Frequency and dissipation vs. time of a typical modified PEI/PSS multilayer coating. For clarity only 3rd, 5th and 7th overtones are shown. A similar plot for the unmodified PEI/PSS pair can be found in the supporting materials.

399

400 *3.2. PEM stability*

401 Stability of the PEM at high salinities was studied in four cases: PEMs 402 prepared from modified and unmodified PEI and with positive or negative



Figure 5: Evolution of thickness of layers as evaluated by QCM-D. Case of negative capping layer. The left axis represents the mass of the absorbed film per square cm of sensor. The right axis represents approximate thickness of the film based on the assumption of a density of $1.2 \ gr/cm^2$. Error bars represent standard error.

⁴⁰³ capping layer (final layer). These experiments were time consuming due ⁴⁰⁴ to the long times needed to reach the constant signal, and the many steps ⁴⁰⁵ involved (more than 72 hours for modified PEI/PSS PEMs). This rendered ⁴⁰⁶ processing of the data with D-find software computationally intensive and ⁴⁰⁷ not possible. As a result, change of the frequency (7th overtone) was followed ⁴⁰⁸ as a rough indicator of the adsorbed mass of the film. Changes in frequency were normalized by using the equation 1 wherein f_0 denotes the frequency of the baseline.

4



$$\Delta f/f_0 = \frac{-(f - f_0)}{f_0} * 100\% \tag{1}$$

Figure 6: Percentage mass loss after contact with the saline solution versus salinity of the medium. Case of the negatively capped films (4 layers). Frequencies measured after reaching stable signals ($\Delta f/t < 0.1 Hz/min$). Error bars represent standard error from 4 samples. Some of the steps are not reported to enhance the clarity of the chart.

The frequency shifts observed here are the result of two simultaneous phenomena:



Figure 7: Percentage mass loss after contact with the saline solution versus salinity of the medium. Case of the positively capped films(5 layers). Frequencies measured after reaching stable signals ($\Delta f/t < 0.1 Hz/min$). Error bars represent standard error from 3 samples. Some of the steps are not reported to enhance the clarity of the chart.

I. Poly electrolyte detachment due to the instability,

414 II. Film water content due to the internal salt content of the film;

Although the comparison was made between PEMs in Milli-Q water environment, a portion of the Na⁺ and Cl⁻ ions absorbed by the PEM during the contact with the saline solutions could not diffuse out during the rinsing and there are slight differences in internal salinity of the film. This results in the increased swelling and increased water content of the PEM. Detachment of the polyelectrolytes from the PEM (mechanism I) results in an increase in the frequency, whereas higher water content of the film results in a reduction in the frequency. As such I and II can act against each other. The final observed frequency is a summation of effects from I and II.

424 3.2.1. Negative capping layer

PEMs formed in four layers are ending in PSS and thus possess a negative 425 surface charge. A summary of results for these PEMs is available in Fig. 426 6. Considering the sensitivity of the analysis and the error values, It can 427 be claimed that the PEMs formed from unmodified PEI fully retained their 428 stability until the salinity of 4.75M. A more or less stable film results in 429 small absolute values for mechanism I which are comparable to absolute 430 values of frequency shift due to II. Thus the summation of I and II results 431 in small changes resonating around the zero value. The situation was more 432 complex for the case of the modified PEI/PSS film. An initial mass loss was 433 observed on the first contact with saline medium, whereby up to 20% of the 434 PEM mass was lost. However, the multilayer becomes more stable after this 435 initial loss and retains most of its weight until reaching salinity of 4.75M 436 NaCl. 437

438 3.2.2. Positive capping layer

Five layered PEMs end in polycation and possess positive surface charge. Results from this set of experiments are reflected in Fig. 7. Here, in both cases of modified and unmodified polycations we observe less stable films. In the case of unmodified PEI, an initial mass loss of around 25% is observed. This loss increases to up to 60% by reaching 1.5M and up to 75% by reaching 1.75M. Further increase in the salinity doesn't result in considerable mass
loss (refer to Fig. 7). The 5 layer PE films are clearly unstable and extremely
susceptible to salinity. Furthermore, the stepwise loss in the mass suggests
removal of the layers from the film and conforms with the quasi layered
(versus completely mixed) structure.

To best explain the observed behaviors, it's helpful to consider the "in 449 and out" mechanism of LBL deposition as articulated by Richert et al. [55] 450 and discussed in [56]. Observed results can be explained by the relative mo-451 bility of the three PEs involved. Polymer mobility is a complex function of 452 the rigidity, size, pH, linearity, etc. [56]. However, it is known that polymer 453 diffusion is roughly inversely proportional to its molecular weight [57]. The 454 huge difference in the molecular weight of the PEI and PSS used in this 455 study insures lower mobility of the polycation. This is further reinforced 456 with highly branched structure of the PEI. Modified PEIs, possess rela-457 tively spacious crownether pendants that reduce their mobility even further. 458 Moreover, crownethers can form hydrogen bonds with protonated amines 459 [15] which can act as weak cross linking in PEI chains and further reduce 460 the mobility of the modified polycation and hinder PSS diffusion into them. 461 Consequently, the order of mobility of PEs would be PSS>PEI>modified 462 PEI. As such, we expect low degree of intermixing between layers in the case 463 of both modified and unmodified PEMs, with the modified PEMs being even 464 less intermixed than their unmodified counterparts. The results of the ex-465 periment (linear growth, refer to section 3.1) conform with this expectations 466 as well. Moreover, this explanation is in agreement with the molecular dy-467 namics simulation and experimental results of Lee et al. [58] who found 468 PEM films made from bigger polymers more layered and suggested a similar 469 explanation for it. 470

As mentioned by the [56], in these situations the more mobile polymer 471 diffuses in the film while the less mobile species retains its more or less 472 layered structure. We hypothesize that the PSS chains are diffusing inside 473 the previously deposited layers and by complexing with PEI act like a cross 474 linker between previous PE chains. In the absence of final capping PSS layer, 475 the links between top poly cation chains (5th layer) and rest of the film is 476 exceptionally weak. This effect is more pronounced in the case of modified 477 PEIs and explains the less stable PEMs with positive capping layer. 478

Once the inadequately anchored 5th layer is removed, an interesting case arises from comparing with the 4-layered membranes. It appears that upon addition of the less mobile PEI layer, stability of the previous layers is also compromised. However this is expected in lights of the "in and out" model. Upon addition of the PEI layer, PSS chains present in the lower layers diffuse towards the surface, weakening the internal links of the first four layers.

Differences observed between the modified and unmodified cases of 4layered PEMs can be explained in a similar fashion. Lower mobility of the modified PEI reduces the relative mobility of the polymer chains and hinders PSS. This reduces the stability, but is not as severe as the case of five layered membranes and once the top layers lose their inadequately anchored polyelectrolytes, the PEM becomes stable.

In both modified and unmodified PEMs, a complete dissolution of the PEM structure was not achievable by increasing salinity and a 2% solution of Hellmanex[®]III solution (Hellma GmbH, Germany) providing a pH of 12 had to be used to completely remove the adsorbed films.

495 3.3. Filtration experiments

Salt rejection experiments were performed with both types of membranes
 and selectivity between different cations was followed as

$$\alpha_{\rm Li/K} = \frac{\frac{C_{\rm Li,permeate}}{C_{\rm Li,feed}}}{\frac{C_{\rm K,permeate}}{C_{\rm K,feed}}}$$
(2)

Wherein C_x denotes concentration of species x in ppm. The permeability of the membrane towards water and different solutes was followed by water and solute permeability (A and B) constants in Solution-Diffusion model [59]. Corrections for the concentration polarization phenomena were made based on the Film theory [60].

$$J_{\rm w} = A(\Delta P - \Delta \pi) \tag{3}$$

$$\frac{C_{\rm m} - C_{\rm p}}{C_{\rm b} - C_{\rm p}} = \frac{\frac{R_{\rm real}}{1 - R_{\rm real}}}{\frac{R_{\rm obs}}{1 - R_{\rm obs}}} = exp(\frac{J_{\rm w}}{k}) \tag{4}$$

$$B = \frac{J_{\rm s}}{C_{\rm m} - C_{\rm p}} = \frac{J_{\rm w}(1 - R_{\rm obs})}{R_{\rm obs} \ exp(\frac{J_{\rm w}}{k})}$$
(5)

Wherein $J_{\rm w}$ and $J_{\rm s}$ denote water and solute flux. ΔP and $\Delta \pi$ represent applied pressure difference and osmotic pressure difference. $C_{\rm p}, C_{\rm b}$ and $C_{\rm m}$ stand for concentration of the solute at the permeate, feed and membrane surface respectively. And finally $R_{\rm obs}$ and $R_{\rm real}$ denote observed and real rejections. The mass transfer coefficient, k, was calculated as: [61]

$$Sh = kr/D = 0.023Re^{0.567}Sc^{0.33}$$
(6)

$$Re = \omega r^2 / \nu \tag{7}$$

⁵⁰³ Wherein Sh, Re and Sc denote Sherwood, Reynolds and Schmidt numbers ⁵⁰⁴ and ω and r represent the angular velocity and radius of the stirrer. The diffusion coefficient, D, was calculated for the feed concentration of the salts
according to the equations provided by Lobo [62].

507 3.3.1. Behavior in the transient state

In experiments with modified membranes performance of the membrane 508 in terms of solute rejection was a function of time, allowing for two stages to 509 be differentiated. At the start of the experiment a transient step could be 510 identified, represented by the gray zone in Fig.8. After about 90 minutes, 511 a plateau was reached and the performance of the membrane was stabi-512 lized. In the case of membranes prepared from unmodified PEI, the value 513 of $\alpha_{\rm Li/K}$ was constant and close to that of the feed. Therefore, the quasi 514 breakthrough curve observed in the transient step can be associated with 515 the complexation of cations and crown ether moieties. We postulate that 516 at around 90 minutes, all crownether sites are saturated and selectivity of 517 the membrane is lost. A similar behavior and explanation is also reported 518 by Tas et al. in crownether containing ion-exchange membranes [42]. It is 519 worth mentioning that regenerated membranes exhibit comparable behavior 520 over several regeneration cycles and as such, the loss of selectivity can not 521 be associated with the instability of the membranes. 522

The modified membranes show a tendency towards hindering the lithium 523 more than potassium during the transient step, reflected in $\alpha_{\rm Li/K}$ values of 524 lower than one. This points to a stronger interaction between lithium ions 525 and the crown ether in comparison to the potassium cations. Although the 526 order of preference of free crown ether in the bulk is in reverse (refer to 527 Table 1), a change in the complexation was expected in crown ethers inside 528 the membrane pores. Potassium is generally believed to form sandwich type 529 2:1 complexes with the 15-crown-5 [23]. A complex type that is severely 530



Figure 8: $\alpha_{\text{Li/K}}$ as a function of time

hampered due to the steric hindrance in highly branched structure of the
PEI. In contrast, lithium ions are smaller and form 1:1 complexes. We
hypothesize that the effect of steric hindrance has reduced the possibility of
2:1 complex formation for potassium, leading to stronger lithium complexes
in comparison to potassium ones.

It is worth mentioning that each crownether molecule has been in contact with several cations before it reaches the saturation point. By rough estimations, close to 13 nmol of crownether are present in membrane coupons (refer to the supporting materials for detailed calculation of this value). In



Figure 9: Cation concentration in the permeate as a function of the volume of the permeate. Experiment performed on virgin modified membrane and mixed salt conditions.

a typical sample of regenerated membranes in mixed salt feed, about 11 540 μ mol lithium and 5 μ mol potassium have passed through the membrane 541 before reaching the 90 minute mark. As a result, it can be expected that 542 each crown ether molecule undergoes several complexation/decomplexation 543 cycles before reaching the saturation point. This suggests a kinetic factor 544 that limits the release of the complexed ions, leading to the eventual satu-545 ration and exhaustion of all crownether sites. However the exact reason and 546 mechanism behind the saturation of complexing sites remains elusive and 547

⁵⁴⁸ further investigations are needed to validate this claim.

549 3.3.2. Behavior in steady state

In general, adding the crownether molecules to the PEI chain resulted 550 in membranes with lowered permeability and increased salt rejection. Upon 551 PEI modification, water permeability constant (A) was lowered by close to 552 70% from 4.53 E-7 to 1.26 E-7 $\frac{m}{s\ bar}.$ At the same time, the salt rejection of 553 the membranes increased from a value of 6.8% to 71.6% for the case of KCl 554 and 3.5% to 41.1% in the case of LiCl in crownether containing coupons. 555 This is in accordance with the changes as big as one order of magnitude in the 556 solute permeability (B) constants (refer to table 3). While such a behavior 557 is in agreement with the general permeability/selectivity trade-off [63], its 558 nature should be further investigated. One of the possible explanations 559 can be the increase in the thickness of the modified membranes. In order 560 to verify this hypothesis, a comparison between 6 layer (3 bilayer) and 14 561 layer (7 bilayer) membranes was made (for a detailed explanation of the 562 synthesis of 14 layer membranes and the performed experiment please refer 563 to the supporting information). While the unmodified 14 layer membrane is 564 thicker than its 6 layer modified counterpart, it still shows a lower rejection. 565 Therefore, increased thickness can not be assumed the sole reason for the 566 increased rejections. 567

Since the lack of cation selectivity in this step rules out any effect from complexation, we postulate that a structural difference between the two PEMMs is responsible for the increased rejection. The structural difference can be caused by the change in mobility and hydrophobicity of the PEI upon modification and is also suggested by the QCM-D results of Sec. 2.2 and Sec. 3.2. For example, a more layered structure is expected to increase Table 3: Water and solute permeability constants A and B for single salt experiments at steady state.

Membrane	$\mathbf{A}(\frac{m}{s \ bar})$	$\mathbf{B_{Li}}\ (m/s)$	$\mathbf{B}_{\mathbf{K}}(m/s)$
Unmodified PEI/PSS	4.53 ± 0.49 E-7	7.0 ± 0.03 E-8	6.8 ± 2.4 E-8
modified PEI/PSS	1.26 ± 0.08 E-7	1.4 ± 0.1 E-8	9 ± 2 E-9

rejection. This will be the subject of a future study. It is worth mentioning that at steady state, the solute permeability factor (B) of lithium is 1.5 times higher that of one of the potassium. However, this is not in contradiction to the results of the transient state (α values). The crown ether sites are all saturated at this stage and do not affect the ion transportation. The higher permeability of lithium in this stage can be contributed to its higher intrinsic mobility in the membrane.



Figure 10: Comparison of Rejection values for Li and K in single salt experiments at steady state step.

581 3.4. Applicability

The goal of this study was to achieve a lithium reach permeate stream by manufacturing a monovalent selective PEMM membrane. However, the resulted membranes exhibited a potassium selective nature. Nevertheless, we believe the results of the study can contribute to the goal of the study as a proof of concept.

Cation selectivity of the PEMM membranes with embedded crown-ether 587 moleties is proven in this study. The authors postulate that replacing the 588 chosen crown-ether with other ligands of this family results in membranes 589 with different cation selectivities. It should be noted that the synthesis 590 route of the crown ether moiety (section 2.1.1) is not exclusive to the 15-591 crown-5. Different crown ethers can be synthesized via the same route by 592 changing the linear ether precursor (here bis[2-(2-chloroethoxy)ethyl]ether). 593 The presented synthesis procedure can thus be used for other cation/crown 594 ether pairs as well. Selection of the appropriate crownether for the future 595 studies can be based on the reasoning provided in this manuscript. 596

597 4. Conclusion

The crown ether molecule 15-crown-5 was synthesized and incorporated in the PEI chains. The modified PEI polymers were used as one of the poly electrolytes in LBL deposition in conjugation with PSS as poly anion. The stability of the PEM was established on silica substrate and the PEMMs were prepared and characterized in terms of water, lithium and potassium permeability. The synthesized membranes showed selectivity only for a defined period of time.

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